

Why grain boundaries are always detrimental for the device performance of solar cells and dislocations are not (in some material systems)

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In solar-cell devices, grain boundaries reduce the device performance mainly via enhanced nonradiative recombination, leading to losses in the open-circuit voltage ( $V_{oc}$ ). As discussed in a recent report [1], the barriers for charge carriers at grain boundaries in all relevant photovoltaic materials - Si, CdTe, Cu(In,Ga)Se<sub>2</sub>, Cu<sub>2</sub>ZnSn(S,Se)<sub>4</sub>, halide perovskites - do not exhibit values sufficiently large (only several  $\pm 10$  meV) to affect charge transport substantially and thus, to affect the current of the solar-cell device. After analyzing grain boundaries in more than 35 solar-cell absorbers based on various material systems (i.e., the ones mentioned above), it can be concluded that grain boundaries influence the conversion efficiency mainly via decreases in  $V_{oc}$  and the corresponding part of the fill factor linked to the  $V_{oc}$ . The  $V_{oc}$  losses due to recombination of charge carriers at grain boundaries can be estimated using a simple model [2,3], ranging from few mV to few 100 mV, depending on the absorber material. Thus, grain boundaries in solar-cell absorbers are always detrimental for the device performance.

At dislocations in Si and CdTe, enhanced nonradiative recombination occurs in a similar manner as at grain boundaries in the same material. However, this is not the case for Cu(In,Ga)Se<sub>2</sub>, in which luminescence signals measured at dislocations do not differ from those acquired in the surrounding matrix [4]. I will discuss the reason for this behavior and give an overview of a microscopic study analyzing epitaxial Cu(In,Ga)Se<sub>2</sub> layers with various composition and with/without the addition of Na. It will be shown that dislocations in Cu(In,Ga)Se<sub>2</sub>, independent of the composition, do not exhibit different luminescence emissions than the matrix and thus, cannot be expected to affect the  $V_{oc}$  values of the solar cells considerably, in contrast to grain boundaries in the same material. I will outline why a similar behavior can be expected for dislocations in Cu<sub>2</sub>ZnSn(S,Se)<sub>4</sub> and halide perovskites.

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[2] D. Abou-Ras, S. Weitz, J. Huang, K. Sun, Y. Gong, A. Jimenez-Arguijo, M. Dimitrievska, X. Hao, E. Saucedo, A comprehensive analysis of recombination at grain boundaries in high-efficiency kesterite-type solar cells, to be published in *Energy Environ. Mater.* (2025), doi: 10.1002/eem2.70048.

[3] L. Blazevic, S. Weitz, E. Artegiani, A. Romeo, C.-Y. Song, J. Horstmann, D. Abou-Ras, Analyses of recombination velocities at grain boundaries by cathodoluminescence: control of injection level and effect of grain-boundary inclination angle, submitted to *Nanotechnology* (2025).

[4] D. Abou-Ras, A. Nikolaeva, M. Krause, L. Korte, H. Stange, R. Mainz, E. Simsek Sanli, P.A. van Aken, T. Sugaya, J. Nishinaga, Optoelectronic inactivity of dislocations in Cu(In,Ga)Se<sub>2</sub> thin films, *phys. stat. sol. (RRL)* 15 (2021) 2100042, doi: 10.1002/pssr.202100042